

Fullerene/Porphyrin Multicomponent Nanostructures on Ag(110): From Supramolecular Self-Assembly to Extended Copolymers

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At present, self-assembly of organic molecules on different substrates is one of the most studied approaches in order to obtain surface-supported supramolecular nanostructures with controlled dimensions and innovative properties. This bottom-up strategy can potentially become a simple, versatile, and effective approach to obtain predetermined and defect-free nanostructured two-dimensional (2D) macroscopic materials. Its success depends on our capacity to build “smart” molecular building units, able to organize themselves in a predetermined pattern by a careful exploitation of intermolecular and molecule–substrate interactions. To this end, only a detailed study of the physicochemical properties at the atomic and molecular scale of the self-organizing interfacial systems can provide the necessary know-how to build increasingly complex and chemically stable 2D functional nanosystems in a reproducible way.^{1–3}

A very large number of complex two-dimensional nanonetworks has been obtained on different single crystal substrates by deposition of organic molecules whose self-assembly is driven by subtle equilibria between weak and reversible lateral (intermolecular) and vertical (molecule–substrate) interactions.⁴ Specifically, dispersion and van der Waals forces,⁵ dipole–dipole and donor–acceptor interactions,^{6,7} hydrogen bonds,^{8,9} or metal complexation¹⁰ allow for good molecular mobility on the surface and, as a consequence, promote the formation of target structures under equilibrium conditions, which provide the apt environment for the

ABSTRACT A novel two-step bottom-up approach to construct a 2D long-range ordered, covalently bonded fullerene/porphyrin binary nanostructure is presented: in the first place, reversible supramolecular interactions between C₆₀ and 5,15-bis(4-aminophenyl)-10,20-diphenylporphyrin are exploited to obtain large domains of an ordered binary network, subsequently a reaction between fullerene molecules and the amino-groups residing on porphyrin units, triggered by thermal treatment, is used to freeze the supramolecular nanostructure with covalent bonds. The resulting nanostructure resists high temperature treatments as expected for an extended covalent network, whereas very similar fullerene/porphyrin nanostructures held together only by weak interactions are disrupted upon annealing at the same or at lower temperatures.

KEYWORDS: fullerene · porphyrins · silver · multicomponent surface self-assembly · covalent molecular networks · STM

necessary self-correction of defects, ultimately resulting in well-developed long-range order. On the other hand, these types of nanostructures are intrinsically fragile and do not resist more aggressive conditions such as high temperature thermal annealing or air exposure, they have little or no mechanical stability, and do not provide efficient intermolecular charge transport.

A nanostructure held together by strong and irreversible covalent bonds can be a solution in order to overcome these limitations.¹¹ Indeed, in the very past few years, different surface science groups have spent much effort in order to study the formation of covalent bonds between surface-supported organic molecules to obtain more complex molecules,^{12–15} monodimensional or bidimensional ordered oligomers,^{15–18} and more extended network structures.^{19,20} Nevertheless, as well evidenced by Perekpichka and Rosei,²¹ so far the dimensions and the long-range order of these polymer nanostructures are low with respect to the arrays self-assembled through weak supramolecular interactions.

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Received for review May 25, 2010 and accepted August 06, 2010.

Published online August 13, 2010.
10.1021/nn101161a

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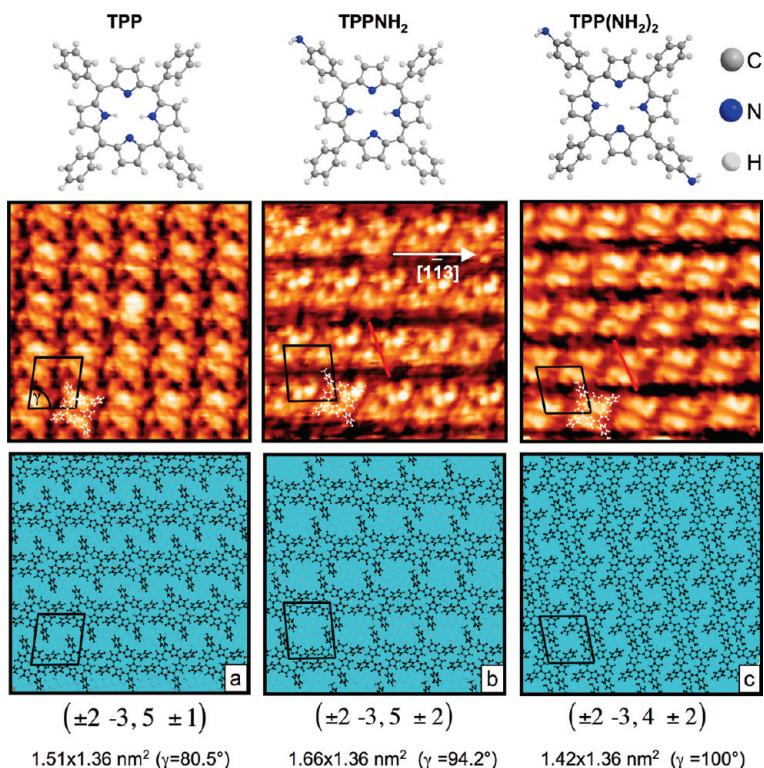


Figure 1. Single molecule models (top), high resolution STM images (center), and related superstructure models (bottom) of three different porphyrins on Ag(110): (a) TPP ($7 \times 7 \text{ nm}^2$, $V = 0.53 \text{ V}$, $I = 3.2 \text{ nA}$); (b) TPPNH₂ ($7 \times 7 \text{ nm}^2$, $V = 0.03 \text{ V}$, $I = 0.03 \text{ nA}$) and (c) TPP(NH₂)₂ ($7 \times 7 \text{ nm}^2$, $V = -0.22 \text{ V}$, $I = 2.27 \text{ nA}$). A molecular model (white) is superimposed to a single molecule in each STM image. The Ag(110) substrate atoms are shown as light blue circles in the bottom-most row. Unit cells are drawn in black. For every unit cell, the matrix notation with respect to the substrate and unit cell dimensions (unit vector lengths and angle γ between unit vectors) are reported. Red lines in panels b and c highlight the most probable direction of alignment of aminophenyl rings.

One of the main reasons for the high defectivity is the low activation temperature for covalent bonds formation between the molecular building units that leads to the rapid growth of large polymolecular fragments with low mobility, in turn hindering the attainment of an equilibrium ordered structure.

A possible strategy in order to obtain a large 2D long-range ordered, covalently bonded nanostructure is a two step process: one should first obtain a large 2D well-ordered supramolecular architecture exploiting weak interactions between the organic molecular modules and then stabilize this structure with covalent bonds by a thermal or phototreatment able to trigger a specific reaction between the building units which does not occur spontaneously during the self-assembly process. Within this approach, it is important that initial weak and final covalent interactions link molecules in a similar way, that is, the preorganization step should lead to a supramolecular network wherein the molecular modules are linked to each other in a such a way that the subsequent covalent bond formation implies minimal rearrangements in the superstructure, since substantial repositioning inevitably leads to a high level of disorder.^{16–18}

This paper describes the advances made in our laboratory toward a bottom-up fabrication of a 2D long-range ordered, covalently bonded organic binary nano-

structure composed of 5,15-bis(4-aminophenyl)-10,20-diphenylporphyrin (TPP(NH₂)₂ molecule in Figure 1) and C₆₀ on Ag(110). This nanostructure has been obtained by means of a two-step strategy: in the first place weak, reversible interactions between C₆₀ and TPP(NH₂)₂ are exploited to obtain very large domains of an ordered binary network. Subsequently a reaction between the amino-groups of TPP(NH₂)₂ and the fullerene molecules triggered by thermal treatment is used to fix the supramolecular nanostructure with covalent bonds.^{22–25} All the experiments have been carried out in ultrahigh vacuum (UHV) conditions and the scanning tunneling microscopy (STM) analysis of the surface was performed at room temperature (RT). Sample bias values are reported throughout the article.

RESULTS AND DISCUSSION

As summarized in Figure 1, three types of porphyrin have been investigated: simple tetraphenylporphyrin (TPP), 5-(4-aminophenyl)-10,15,20-triphenylporphyrin (TPPNH₂), where one phenyl group has been substituted by a *p*-aminophenyl group, and TPP(NH₂)₂, where two *trans* phenyl groups have been substituted by two *p*-aminophenyl groups. In this article the general acronym TPP collectively indicates all three tetraphenylporphyrin species when no ambiguity arises from this usage.

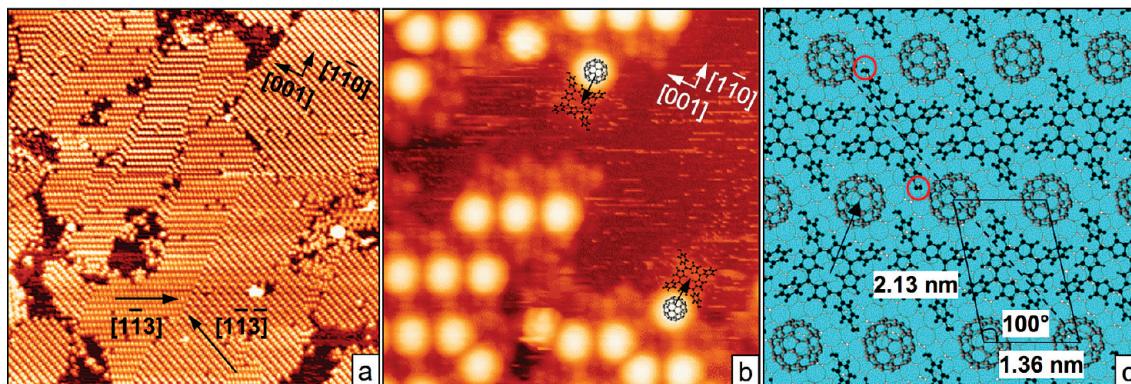


Figure 2. ($\pm 2 - 3, 6 \pm 3$) bicomponent $\text{TPP}(\text{NH}_2)_2/\text{C}_60$ self-assembled nanostructure on $\text{Ag}(110)$ obtained by annealing the system at 410 K and measured at RT. (a) Large scale STM image ($140 \times 140 \text{ nm}^2$, $V = -0.4 \text{ V}$, $I = 4.0 \text{ nA}$). (b) High resolution STM image ($15 \times 15 \text{ nm}^2$, $V = -0.3 \text{ V}$, $I = 0.2 \text{ nA}$) of a defective domain boundary: molecular models are superimposed to two $\text{TPP}(\text{NH}_2)_2/\text{C}_60$ couples; arrows symbolize the weak lateral intermolecular interactions between the two species aligned along the substrate $[1\bar{1}0]$ direction. (c) Molecular model of the bicomponent nanostructure. The $(2 - 3, 6 3)$ superstructure unit cell is shown. Red circles highlight $-\text{NH}_2$ groups on a $\text{TPP}(\text{NH}_2)_2$ molecule.

The central and bottom rows of Figure 1 show STM images and molecular models, respectively, of TPP, TPPNH₂, and TPP(NH₂)₂ monolayer islands deposited on Ag(110) at RT. For symmetry reasons, each molecular species self-organizes in two equivalent domains rotated by 59° from each other. All overlayers form simply commensurate superstructures (integer matrix elements in matrix notation), which indicates that all molecules in each overlayer occupy equivalent adsorption sites. A tentative overlayer-substrate registry is sketched in Figure 1, bottom. From comparison of the three different unit cells, it is evident that the cell parameter aligned along the $[1\bar{1}3]$ substrate direction is common to the three TPPs, whereas the second cell parameter shows different lengths and directions depending on the molecule. Therefore, on changing the number and the position of aminophenyl rings, the molecular stripe direction and the TPP-TPP nearest neighbor (NN) distance along the $[1\bar{1}3]$ -oriented stripes do not change (which is reflected in the invariance of the first row in matrix notations for the three superstructures), whereas the interstripe separation is modified, giving rise to different surface densities for the three phases. This is an indication that the aminophenyl rings of TPPNH₂ and TPP(NH₂)₂ point to adjacent stripes rather than to nearest neighbor molecules within each $[1\bar{1}3]$ -aligned stripe. (i.e., they are aligned along the red lines in the STM images of Figure 1). Intrastripe orientation of aminophenyl rings is unlikely also for steric reasons, as already discussed in detail elsewhere.²⁶

Codeposition of TPP and C₆₀ submonolayers on the Ag(110) substrate held at RT has been optimized in order to obtain an approximately 1:1 ratio between the two components (as already described in ref 26) which leads to the formation of a short-range ordered bicomponent nanostructure organized in small islands.

After extensive annealing (~400 min) at 410 K, molecules self-assemble in a very large 2D long-range or-

dered and commensurate nanostructure, resulting from the equilibrium between weak reversible lateral TPP-C₆₀ and vertical molecules–substrate interactions. Figure 2 displays STM results concerning TPP(NH₂)₂. The large scale STM image reported in Figure 2a shows very extended, long-range ordered and low defectivity islands (again two domains rotated by 59° from each other form for symmetry reasons), wherein single domains as large as 7000 nm^2 can be observed.

This complex nanostructure is strictly equivalent to the TPPNH₂/C₆₀ supramolecular network described in detail elsewhere.²⁶ This is characterized by alternating stripes of C₆₀ and TPPNH₂ aligned along the two $[1\bar{1}3]$ and $[1\bar{1}\bar{3}]$ equivalent substrate directions (see Figure 2a) forming the commensurate unit cell sketched in Figure 2c, where the two species appear in a 1:1 ratio and are both in direct contact with the substrate. The C₆₀–C₆₀ NN distance along a stripe is $1.4 \pm 0.1 \text{ nm}$, that is, it mirrors the intrastripe NN intermolecular distance found in the pure porphyrin phase and is substantially larger than the NN interfullerene distance (1.0 nm) in the close-packed $c(4 \times 4)$ phase produced by fullerene alone on Ag (110).²⁷

The same bicomponent superstructure has been obtained by self-assembling C₆₀ with all the TPPs reported in Figure 1. We conclude that the weak interactions responsible for the self-organization of this supramolecular structure are not influenced by the presence and the number of peripheral amino groups on the TPP molecules, at least in the range that has been explored. From high resolution STM images collected at the borders of ordered domains, where several isolated C₆₀–TPP couples can be singled out, it is possible to infer the orientation of this weak TPP–C₆₀ interaction: in Figure 2b two C₆₀ and two TPP(NH₂)₂ molecules have been evidenced with the respective molecular models in order to indicate that every TPP(NH₂)₂ unit “captures” a C₆₀ molecule along the substrate $[1\bar{1}0]$ direction (marked with black arrows) by laterally coordinating it,

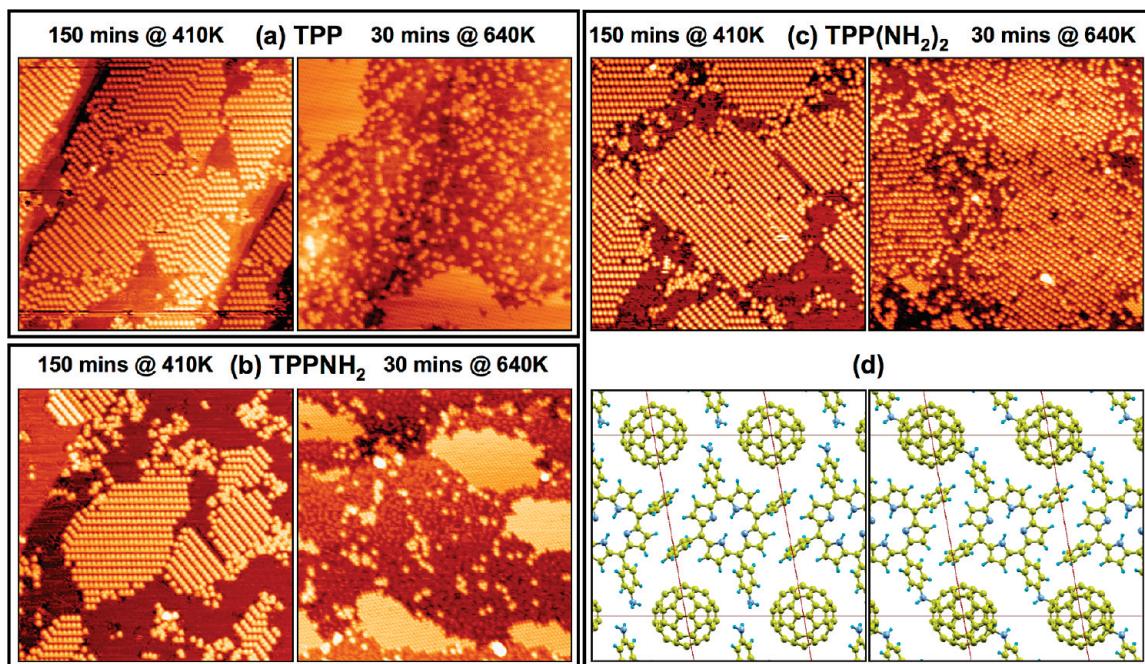


Figure 3. Different thermal stability of binary nanostructures self-assembled starting from C_{60} and (a) TPP ((left) $V = -0.8$ V, $I = 1.0$ nA; (right) $V = -0.7$ V, $I = 1.0$ nA); (b) TPPNH₂, ((left) $V = 0.5$ V, $I = 0.9$ nA; (right) $V = 0.8$ V, $I = 1.9$ nA), (c) TPP(NH₂)₂ ((left) $V = -0.5$ V, $I = 0.3$ nA; (right) $V = -0.5$ V, $I = 0.4$ nA). All STM images are 85×85 nm² in size. Annealing temperatures and times are indicated. (d) Molecular model of the supramolecular network (left) and a tentative model of the covalently linked structure (right).

possibly through a couple of adjacent peripheral phenyl rings linked to the porphyrin macrocycle. Time-lapsed STM imaging confirms the reversibility of such supramolecular interactions at the borders of the already formed bicomponent islands.

As evidenced by red circles in Figure 2c, the two aminophenyl rings of each TPP(NH₂)₂ molecule point toward NN C_{60} molecules rather than toward the NN TPP(NH₂)₂ units belonging to the same TPP stripe. Such an arrangement is required for steric reasons,²⁶ and it is also deduced by comparing the unit cells of the different single component porphyrin ordered phases on Ag(110) reported in Figure 1 to the unit cell of the bicomponent phase: as already remarked, the superstructure unit cell parameter along the [113] substrate direction is preserved (1.4 nm) on going from the single component overlayers to the bicomponent supramolecular network. In addition, the proposed orientation of aminophenyl rings is a prerequisite for the polymerization reaction discussed below.

To this point, the behavior of the three different TPP molecules with respect to C_{60} is exactly the same when the sample is annealed at 410 K. However, the thermal stability of this common superstructure is very high when TPP(NH₂)₂ is used as a building unit in the bicomponent self-assembly, while it is definitely lower when either TPP or TPPNH₂ are employed. The left columns of Figure 3 show the common nanostructure obtained using the three different porphyrins by means of a long annealing treatment (150 min) at 410 K, whereas the right column reports images of the result-

ing surface morphology after an additional annealing treatment of 30 min at 640 K. It appears that when one starts from TPP and TPPNH₂ the bicomponent ordered network is disrupted upon high temperature annealing. Phase separation occurs and consequently most C_{60} molecules segregate in close packed c(4 \times 4) domains,²⁷ while porphyrin molecules form a disordered phase.^{26,28} On the other hand, the nanostructure self-assembled starting from TPP(NH₂)₂ resists the thermal treatment: the local order and the dimensions of the ordered islands remain approximately unaltered. The observed high thermal stability is in line with the behavior of other covalent organic frameworks^{19,20} and its high temperature limit seems to be determined by the onset of TPP degradation at about 680 K, as obtained from thermogravimetric analysis (see Supporting Information).

It is evident that the different response to the high-temperature thermal treatment strictly depends on the presence and number of amino groups on TPP molecules. The ability of amines to react with C_{60} both in self-assembled monolayers in solution^{22–24} and in the solid state is well established: this property has been exploited for the production of fullerene polymers, as recently reviewed.²⁹ The model sketched in Figure 3d shows that the two trans amino groups on each TPP(NH₂)₂ molecule are very close to two C_{60} units positioned on adjacent stripes: with a few degrees of anti-clockwise rotation and/or with a little shift of TPP(NH₂)₂ monomers, the two amino groups can be positioned at a covalent bonding distance with respect to both ad-

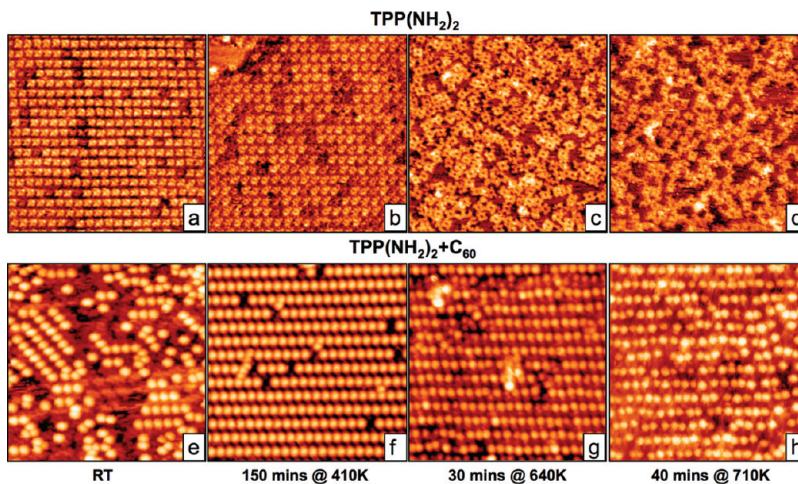


Figure 4. Thermal stability of the single component $\text{TPP}(\text{NH}_2)_2$ self-assembly (a–d, top) compared to the thermal stability of the bicomponent $\text{C}_{60}-(\text{TPP}(\text{NH}_2)_2)$ system (e–h, bottom): (a) $V = -0.4 \text{ V}$, $I = 3.0 \text{ nA}$; (b) $V = 0.65 \text{ V}$, $I = 10.0 \text{ nA}$; (c) $V = -0.24 \text{ V}$, $I = 3.0 \text{ nA}$; (d) $V = -0.53 \text{ V}$, $I = 15.0 \text{ nA}$; (e) $V = -1.0 \text{ V}$, $I = 1.0 \text{ nA}$; (f) $V = -0.55 \text{ V}$, $I = 0.35 \text{ nA}$; (g) $V = 1.0 \text{ V}$, $I = 2.6 \text{ nA}$; (h) $V = -0.20 \text{ V}$, $I = 1.6 \text{ nA}$. All STM images are $15 \times 15 \text{ nm}^2$ in size. Annealing temperatures and times are indicated for each column.

jacent C_{60} molecules, resulting in the copolymer chains sketched in Figure 3d, right, where molecules bind along the longer diagonal of the superstructure unit cell. On the basis of the available STM data, we cannot exclude the other possibility, namely the clockwise rotation of $\text{TPP}(\text{NH}_2)_2$ monomers in order to bind along the shorter diagonal of the oblique unit cell. It is worth noting that if the two bonding schemes were energetically roughly degenerate, a randomly linked 2D covalent network would occur. Quantum chemistry calculations explicitly including the interactions of both molecular species with the substrate are needed to address this interesting issue.

In summary, the peculiar high thermal stability of the binary nanostructure containing $\text{TPP}(\text{NH}_2)_2$ molecules can be explained by the fact that both amino groups on each $\text{TPP}(\text{NH}_2)_2$ unit react with nearby C_{60} molecules in the preorganized supramolecular network forming covalently bonded $(-\text{TPP}-\text{NH}-\text{C}_{60}\text{H}_2-\text{NH}-)_n$ copolymer chains or possibly a 2D network. Very similar water-soluble main-chain fullerene polymers with a molecular weight up to 20.0 kg/mol, wherein C_{60} units are linked through aromatic diamines, have been also reported to form in solution.³⁰

To substantiate the hypothesis of covalent bonding between the molecular building units, the thermal stability of the three TPP molecules has been investigated in order to exclude that a different thermal behavior of TPP, TPPNH₂, and $\text{TPP}(\text{NH}_2)_2 + \text{C}_{60}$ bicomponent nanostructures is due to an intrinsically higher thermal stability of $\text{TPP}(\text{NH}_2)_2$ on Ag(110). Each TPP molecule has been deposited on Ag(110) held at RT. Three similar single component ordered phases (see the example of $\text{TPP}(\text{NH}_2)_2$, Figure 4, top) have been obtained and then annealed at 640 K for 30 min, following the same thermal treatment used for the bicomponent systems (Figure 4 (bottom) shows high resolution close ups of the

$\text{TPP}(\text{NH}_2)_2 + \text{C}_{60}$ system). The resulting single component overlayer has the same characteristics, irrespective of the TPP used: the starting ordered phase is disrupted and the surface density of TPP molecules decreases by approximately 40% judging from STM images of different sample areas, while the C 1s XPS signal intensity decreases approximately 20%.

The two different numbers can be reconciled if one assumes that part of the molecules desorbs (640 K is higher than the temperature used for evaporating TPPs from the Knudsen cell) and part decomposes. In fact, desorption contributes both to STM apparent molecular surface density decrease and C 1s intensity reduction, while decomposition contributes only to the former. (Clear signs of molecular fragmentation are seen in Figure 4d, while no significant porphyrin metallation by inclusion of Ag atoms from the substrate is detected in the whole temperature range.) We conclude that the thermal stability of the three TPP molecules on Ag(110) is the same and therefore the strikingly higher stability of the $\text{TPP}(\text{NH}_2)_2/\text{C}_{60}$ bicomponent system is not related to a higher intrinsic stability of $\text{TPP}(\text{NH}_2)_2$ with respect to either TPP or TPPNH₂ in contact with silver. This is a further indication that the binary nanostructure containing $\text{TPP}(\text{NH}_2)_2$ is stabilized through an extended network of covalent bonds. In fact, the corresponding sequence of STM images reported in Figure 4e–h shows that no significant surface density decrease of the molecular species in the bicomponent system is observed upon heating. The notable effect of annealing between RT and 410 K is the development of large ordered domains from small supramolecular islands (Figure 4e,f). Increasing the annealing temperature to 640 K (Figure 4g) only causes a partial loss of registry of the overlayer rows with respect to the substrate, which can be attributed to small conformational adjustment of the covalently linked chains due to torsional

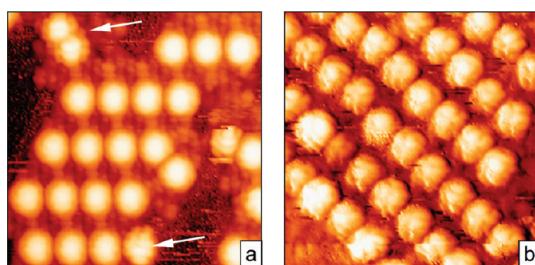


Figure 5. High resolution STM images ($12 \times 12 \text{ nm}^2$), convoluted with their respective current images, of the $\text{TPP}(\text{NH}_2)_2/\text{C}_60$ bicomponent network (a) after annealing at 410 K ($V = 0.3 \text{ V}$, $I = 0.2 \text{ nA}$) and (b) after annealing at 640 K ($V = 1.0 \text{ V}$, $I = 0.5 \text{ nA}$). White arrows indicate fullerene molecules that show intramolecular fine structure at 410 K (see text).

degrees of freedom activated by the thermal treatment. The higher degree of disorder at 710 K (Figure 4h) indicates an incipient decomposition of the overlayer.

A further effect of annealing the bicomponent system at 640 K can be perceived in Figure 3c. Both left and right STM images are acquired at the same bias value. After annealing at higher temperature, C_60 molecules in the network appear less bright than after the thermal treatment at 410 K. Quantitative measurements indeed show that upon annealing there is a $\sim 20\%$ decrease in apparent fullerene height with respect to the substrate in STM images at constant bias (from $5.0 \pm 0.1 \text{ \AA}$ to $4.0 \pm 0.6 \text{ \AA}$ at $V = -0.5 \text{ V}$). No such effect is detected for single component C_60 ad-islands annealed at the same temperature on the same substrate. This observation indicates a redistribution of the density of states on fullerene building units in the $\text{TPP}(\text{NH}_2)_2/\text{C}_60$ bicomponent network which fits the hypothesis of covalent bonding between the building units.

The latter is also supported by high resolution STM measurements where intramolecular fine structure can

be detected. Figure 5 shows two high resolution STM images of the $\text{TPP}(\text{NH}_2)_2/\text{C}_60$ system after annealing at 410 and 640 K.

Both images are convoluted with their respective current images³¹ in order to enhance the contrast of intramolecular fine structure features. While no such features are detected on the majority of fullerenes in the supramolecular network after annealing at 410 K, they are clearly seen after annealing at 640 K on all fullerene units, which indicates that C_60 molecules maintain substantial rotational or at least librational degrees of freedom in the supramolecular network (which is typical for predominantly ionic C_60 —substrate interactions,³² at least as far as C_60 does not form close-packed islands which can stabilize a specific orientation with respect to the surface), while such degrees of freedom are frozen after high temperature annealing, again in accordance with the proposed covalent network formation. It is worth noting that after annealing at 410 K (Figure 5a) a few molecules do show intermolecular fine structure, as indicated by white arrows. The frozen conformation of the two “defective” C_60 molecules in the upper left corner can be attributed to their direct mutual interaction (they are at a distance of approximately 1.0 nm from each other, that is, at the van der Waals NN distance in close packed single component fullerene ad-islands).²⁷ The molecule highlighted in the bottom part of the image belongs to the regular bicomponent network. However, it differs from its siblings for its definite orientation, giving rise to the observed intramolecular fine structure in the STM appearance. These observations indicate (a) that the C_60 — $\text{TPP}(\text{NH}_2)_2$ lateral interactions in the supramolecular network are weaker and less directionally specific than the C_60 — C_60 direct van der Waals interactions in fullerene clusters and close-packed C_60 ad-islands and (b) that at 410 K only a tiny minority of fullerene molecules reacts with $\text{TPP}(\text{NH}_2)_2$ even after extended annealing times, thus setting a lower temperature limit for the occurrence of the covalent bonds formation.

In-situ spectroscopic evidence for the occurrence of the polymerization reaction at 640 K is also provided by X-ray photoelectron spectroscopy (XPS). The N 1s region shows characteristic changes in the course of the reaction, which are in agreement with the proposed bonding scheme.

Figure 6 shows the N 1s XP spectrum of $\text{TPP}(\text{NH}_2)_2$ molecules belonging to the $\text{TPP}(\text{NH}_2)_2/\text{C}_60$ bicomponent network on Ag(110) after annealing at the two relevant temperatures. The spectrum after the thermal treatment at 410 K consists of two components at 400.1 and 398.2 eV, which are assigned to pyrrolic ($-\text{NH}-$) and iminic ($-\text{N}=$) nitrogen, respectively, in accordance with previous studies.^{33–36} In pristine TPP (spectrum not shown) the two components have the same intensity, reflecting the equal number of nitrogen atoms of the two types in the molecule.

In $\text{TPP}(\text{NH}_2)_2$ the high BE component is more intense, since the contribution of the two primary amino

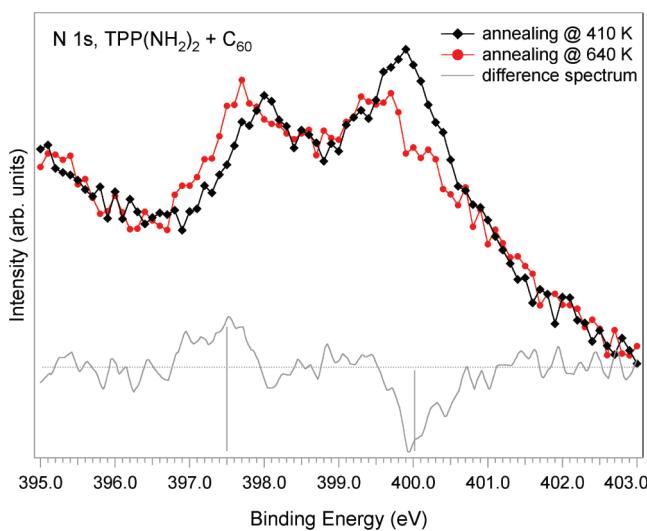


Figure 6. Al K α -excited N 1s XP spectrum of the $\text{TPP}(\text{NH}_2)_2/\text{C}_60$ bicomponent network after annealing at 410 K (black squares) and at 640 K (red circles). Both spectra are referenced to the Ag 3d $5/2$ photoemission peak, whose BE (368.2 eV) does not change upon annealing. The difference spectrum is also shown.

groups (at BE = 400.0 eV)²⁴ cannot be resolved from the pyrrolic contribution. After annealing at 640 K, the high BE component decreases, while a new component appears at 397.5 eV, whose intensity roughly equals the intensity decrease around 400.0 eV (see the difference spectrum in Figure 6). Almost the same binding energy shift is observed in the literature when surface-grafted primary alkylamines react with C₆₀ to form R—NH—C₆₀H adducts.²⁴ We conclude that the XPS spectra of the N 1s region provide a fingerprint for the occurrence of the polymerization reaction.

The obtained covalently linked network is particularly robust and is resistant to air and water exposure (see Supporting Information).

To better understand the occurrence of the fullerene—aminoporphyrin reaction, equimolar amounts of TPPNH₂ and C₆₀ have been mechanically blended with a pestle in an agate mortar. The solid mixture was then heated for 1 h at 85 °C. TLC analysis (petroleum ether/ethyl acetate 3:2 v/v) showed the complete disappearance of the porphyrin ($R_f = 0.56$) and the formation of a new product ($R_f = 0.79$). Both UV-vis and ¹³C NMR analysis (see Supporting Information) confirmed that this new product contained the spectroscopic signatures of the porphyrin and the fullerene units and were different from those obtained for a 1:1 mixture of the two compounds. This result shows that the addition reaction between TPPNH₂ and C₆₀ takes place smoothly even at low temperatures. As shown

here, a substantially higher temperature is required to trigger the reaction in the surface-supported supramolecular aggregate, which is a key for the successful two-step synthesis of the covalently linked nanostructure. A higher reaction temperature on the surface with respect to solution and solid state may be traced back to a substantially reduced entropic term due to frozen orientational degrees of freedom in the supramolecular network, possibly coupled to an active role of the substrate in the form of charge transfer from Ag to C₆₀,^{37,38} which is expected to partially hinder the nucleophilic attack of the amino-groups to fullerene.

In conclusion, we have shown that an extended, ordered, and covalently linked bicomponent molecular superstructure can be obtained by thermal triggering of covalent bonds formation in a preorganized supramolecular network of C₆₀ and TPP(NH₂)₂. These results show promising potential for the synthesis of highly ordered networks of surface-supported functional copolymers. In particular, they point to a possible route for the simple and effective one-pot synthesis of extended regular networks of covalently bonded donor–acceptor (D–A) arrays for efficient charge-separation through the choice of properly functionalized D and A building units. The possibility to extend the polymer conjugation into the second dimension²¹ is currently also investigated through the choice of the fully aminated TPP(NH₂)₄ molecule as a precursor for the bicomponent molecular self-assembly.

METHODS

Synthesis of TPP(NH₂)₂. 5,15-bis(4-aminophenyl)-10,20-diphenylporphyrin was prepared as previously described by Vincente *et al.*³⁹ by nitration of TPPH₂ (purchased from EGA-CHEMIE, Germany) with sodium nitrite in TFA followed by reduction with SnCl₂/HCl. Purification was carried out by column chromatography (Al₂O₃, CH₂Cl₂/petroleum ether 8:2 v/v eluent).

Reaction between TPPNH₂ and C₆₀. A solid mixture of 5-(4-aminophenyl)-10,15,20-triphenylporphyrin and C₆₀ in 1:1 molar ratio was heated at 85 °C. After 1 h, TLC analysis (silica gel, eluent petroleum ether/ethyl acetate 3:2 v/v) confirmed the complete disappearance of TPPNH₂ and the formation of a new product containing both the features of porphyrin and C₆₀, as confirmed by ¹³C NMR and UV-vis spectroscopy (see Supporting Information).

Sample Preparation in UHV. The Ag(110) crystal was cleaned by repeated cycles of 1 keV Ar⁺ sputtering and annealing at 820 K until a clean surface with sufficiently large terraces was confirmed by STM imaging. The three different TPP molecules were deposited from a PBN crucible held at temperatures between 500 and 590 K, whereas C₆₀ (99% purity) was sublimed at temperatures between 850 and 890 K from a tungsten crucible. The Ag specimen was kept at room temperature during sublimations. Both crucibles were outgassed for a long time to avoid impurities during the depositions onto the substrate.

Scanning Tunneling Microscopy (STM). The experiments were performed with an Omicron scanning tunneling microscope (VSTM) operating in UHV at a base pressure of 2×10^{-10} mbar. The STM measurements were carried out at RT in constant current mode, using a Pt–Ir tip. The sample bias voltage (V) and the tunneling current (I) are indicated for all STM images. STM images were analyzed with the WSxM software.⁴⁰

X-ray Photoelectron Spectroscopy (XPS). Measurements were performed *in situ* at RT by means of a Scienta M-780 Al K α (1486.6 eV) monochromatic X-ray source and a Scienta SES-100 photoelectron analyzer fitted to the STM preparation chamber. The

overall instrumental resolution measured at the Fermi edge of the clean Ag sample was 0.3 eV. The binding energies are referenced to the Fermi edge (BE = 0).

Acknowledgment. D. Forrer and A. Vittadini are gratefully acknowledged for useful discussions. This work has been partially funded by MIUR (PRIN 2008, Project n. 2008MXZEAS: “Molecular SPACE”) and by the University of Padova (Progetto Strategico HELIOS and Progetti di Ricerca di Ateneo - CPDA088228/08).

Supporting Information Available: Spectroscopic (¹³C NMR and UV-vis) characterization of the solid state reaction between TPPNH₂ and C₆₀; thermogravimetric analysis of TPP, TPPNH₂ and TPP(NH₂)₂; STM image of air- and water-exposed sample. This material is available free of charge *via* the Internet at <http://pubs.acs.org>.

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